# A Microdischarge-Based Monolithic Pressure Sensor

Christine Kay Eun, Xin Luo, Jun-Chieh Wang, Zhongmin Xiong, Mark Kushner, and Yogesh Gianchandani, *Fellow, IEEE* 

Abstract—This paper describes the investigation of a microdischarge-based approach for sensing the diaphragm deflection in a monolithically fabricated pressure sensor. This transduction approach is appealing from the viewpoint of miniaturization. The device consists of a deflecting Si diaphragm with a sensing cathode, and a glass substrate with an anode and a reference cathode. The total exterior volume of the device is 0.05 mm<sup>3</sup>; typical electrode size and separations are 35 and 10  $\mu$ m. Pulsed microdischarges are initiated in a sealed chamber formed between Si and glass chips, and are filled with Ar gas. External pressure deflects the Si diaphragm and changes the interelectrode spacing, thereby redistributing the current between the anode and two competing cathodes. The differential current is indicative of the diaphragm deflection which is determined by the external pressure. A 6-mask microfabrication process is investigated for device fabrication. Electrode connections to the interior of the chamber are provided by laser drilling and copper electroplating through high aspect ratio glass vias. The Si and glass substrates are bonded by Au-In eutectic. The redistribution of plasma current between competing cathodes, as a consequence of diaphragm deflection over a range of pressure, was experimentally demonstrated. First principles modeling of transient microdischarges have provided insights to the fundamental processes responsible for the differential current and guidance for scaling the device to smaller dimension. [2013-0200]

*Index Terms*—Through-glass vias (TGVs), electroplating, eutectic bonding, plasma modeling.

## I. INTRODUCTION

A VARIETY of microscale pressure sensing solutions have been explored in the past five decades, of which the

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C. K. Eun is with the Center for Wireless Integrated MicroSensing and Systems, University of Michigan, Ann Arbor, MI 48109 USA; and also with the Department of Electrical Engineering and Computer Science, University of Michigan, Ann Arbor, MI 48109 USA (e-mail: eunc@umich.edu).

X. Luo is with the Center for Wireless Integrated MicroSensing and Systems, University of Michigan, Ann Arbor, MI 48109 USA; and also with the Department of Mechanical Engineering, University of Michigan, Ann Arbor, MI 48109 USA (e-mail: xinluo@umich.edu).

J.-C. Wang, Z. Xiong, and M. Kushner are with the Department of Electrical Engineering and Computer Science, University of Michigan, Ann Arbor, MI 48109 USA (e-mail: junchwan@umich.edu; zxiong@umich.edu; mjkush@umich.edu).

Y. Gianchandani is with the Center for Wireless Integrated MicroSensing and Systems, University of Michigan, Ann Arbor, MI 48109 USA; the Department of Electrical Engineering and Computer Science, University of Michigan, Ann Arbor, MI 48109 USA; and also with the Department of Mechanical Engineering, University of Michigan, Ann Arbor, MI 48109 USA (e-mail: yogesh@umich.edu).

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most commonly used are piezoresistive and capacitive pressure sensors [1]. Piezoresistive sensors typically measure stress in a diaphragm as it deflects in response to pressure. In contrast, capacitive pressure sensors respond to diaphragm deflection rather than stress. The smallest micromachined pressure sensors that have been reported – e.g., for use within cardiac catheters – use these transduction techniques [1]. For both types of sensors, the side-dimensions of the diaphragms of the smallest devices have been about 1 mm long. Further reduction in size has been a challenge for both approaches, but for different reasons.

For piezoresistive pressure sensors, reducing the diaphragm diameter presents a challenge in localizing the resistor. If the resistor extends too far from the edge toward the center of the diaphragm, it loses signal due to stress averaging: the stress on the upper surface of the diaphragm changes from tensile at the perimeter to compressive at the center, with a null point located in between. Making the resistor smaller is a challenge as well. Smaller resistors demand more current to generate a measurable voltage, and are relatively imprecise, which affects calibration and yield. Resistors inherently have a high temperature sensitivity, which makes this transduction approach less appealing for high temperature applications. Piezoresistive sensors do have relatively low output impedance, which means that the sensing circuit does not have to be located in the immediate proximity of the sensor. The equivalent noise pressure from piezoresistive pressure sensors increases as  $1/r^4$ , where r is the equivalent radius of the diaphragm [1].

Capacitive pressure sensors present a scaling challenge because the capacitance decreases in proportion to the area of the diaphragm. This scaling puts the burden of detection on the interface circuit, which must not only be precise, but must also be located in the immediate vicinity of the sensor in order to prevent the signal – which comes from a high impedance output, and hence is inherently weak – from leaking into parasitic capacitance. Another consequence of the reduced capacitance is the increase in k<sub>B</sub>T/C thermal noise. Together with other noise sources, the equivalent noise pressure from capacitive pressure sensors increases as  $1/r^5$  [1]. While capacitive pressure sensors have about  $10 \times$  lower sensitivity to temperature than piezoresistive devices, the proximal interface circuit must be tolerant of high temperature environments as well.

Recently, an approach to pressure control and sensing based on microdischarges has been reported [2]–[4] Microdischarges are localized glow discharge plasmas or arcs created in gaseous media, which, due to their size, have characteristics different from larger scale discharges [5]. Microdischarges can be used

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in a variety of microsensors, including micro total analysis systems that use optical emission spectroscopy for chemical sensing [6]-[8], radiation detectors, sputter ion pumps, etc. [9]. Pressure and vacuum sensors based on this method of transduction utilize the change in plasma distribution within a cavity that may be caused either by a change in the plasma pressure [3] or by the deflection of a plasma electrode under external pressure [4]. Sensitivity levels in the range of 40,000-75,000 ppm/kPa have been reported [3], which are comparable to piezoresistive pressure sensors. However, as in capacitive pressure sensors, linearity is compromised. The inherent signal levels are large compared to both capacitive and piezoresistive devices, eliminating the need for a proximal interface circuit - and, in fact, substantially reducing the need for amplification. This is very appealing from the viewpoint of miniaturization. Microdischarge cells that have been reported to date are at least  $100 \times$  smaller in area than the smallest capacitive and piezoresistive pressure sensors that have been reported. For example, the discharge gap in cells used for plasma display panels can be less than 100  $\mu$ m [1]. Microdischarge-based displays with 50  $\times$  50  $\mu$ m<sup>2</sup> cell size have been reported [11]. Emission spectroscopy of microdischarges for portable chemical sensing of liquids and gases has been reported for 50  $\mu$ m gaps between planar metal features [12]. A constraint is that microdischarge-based pressure sensors need high voltage to initiate discharges in the operation.

This paper<sup>1</sup> describes a microdischarge-based approach for the measurement of diaphragm displacements in a monolithically fabricated device. The exterior volume of the device is only 0.05 mm<sup>3</sup>, which is  $\approx 30 \times$  smaller than prior work [4]. The design of the structure is described in Section II, followed by microplasma modeling in Section III. Fabrication processes are addressed in Section IV. Preliminary experiments and results are provided in Section V.

### II. DEVICE CONCEPT AND DESIGN

#### A. Device Concept

The device primarily consists of a glass substrate with copper filled through-glass vias (TGVs), a silicon diaphragm, one anode, and two competing cathodes, as shown in Fig. 1. A microdischarge chamber is formed by the glass substrate, silicon diaphragm, and a Au-In eutectic bond ring. All three electrodes are made of thin-film Ni. The anode (A) and reference cathode (K1) are located on the glass side facing the microdischarge chamber. The sensing cathode (K2) is located on the silicon diaphragm, and is electrically connected to the exterior contact pad through a doped silicon layer and the K2 contact, which is a sandwich of Au and In layers in the interior of the chamber that mates with a TGV. All the electrical connections from within the chamber are routed to the exterior of the glass substrate through copper filled TGVs. An anode electrode placed in the center of the diaphragm would have provided the highest electrode displacement. However, given the diaphragm area available and the location of reference cathode K1, the anode electrode A is offset from the center.



Fig. 1. Concept of the microdischarge based pressure sensor. (a) 3D model of the pressure sensor. (b)  $S-S_0$  view of the structure. *I1* and *I2* are discharge currents from two discharge paths.

In this three-electrode configuration, a voltage pulse is applied to initiate microdischarges, and two current paths are established between the anode and the two cathodes. (The pulsed nature of the microdischarge reduces power consumption and parasitic heating, but requires customized code for simulations and modeling, as discussed later.) As the diaphragm deflects due to external pressure, the spacing between the anode and the sensing cathode (AK2) decreases, but the spacing between the anode and the reference cathode (AKI) is essentially unaffected. This change of interelectrode spacing redistributes the spatial current: the ratio of sensing current (denoted by I2) between AK2, and reference current (denoted by II) between AK1. The differential current, expressed as a fraction of the total peak current (II-I2)/(II+I2), can then be used as the sensor output to indicate the value of external pressure. By using this relative change in current, the absolute current becomes less important, which minimizes the consequence of pulse-to-pulse variation in microdischarge characteristics.

## B. Structural Design

The diaphragm is made of silicon to enable both large diaphragm deflection, within the fracture limit, and electrical conductivity. The eutectic bond ring is 200  $\mu$ m-wide; it forms a sealed chamber (185  $\mu$ m-long, 140  $\mu$ m-wide) and determines the chamber height (and consequently the *AK2* interelectrode spacing). With this approach, etching the

<sup>&</sup>lt;sup>1</sup>Portions of this paper appear in conference abstract form in [13].



Fig. 2. Paschen curves for various fill-gases, which illustrate the relationship between operating voltage and  $p \cdot d$  product, where p is the pressure and d is the interelectrode spacing. Discharge electrodes are Ni. The original data in this plot were reported in [20].

silicon or the glass to form a chamber is avoided. Electrostatic finite element analysis (FEA) using COMSOL® confirms that the use of this conductive bond ring as a spacing layer has little or no impact on the electric field profile. The discharge electrodes are made of nickel, which offers several benefits, including a high secondary electron emission coefficient (that contributes to a lower operation voltage), a high resistance to oxidation compared to alternatives [4], a convenient thin film deposition or electroplating, and ease of patterning. The architecture is designed with the vias in a glass wafer to provide adequate electrical isolation between the electrodes during device operation. This isolation allows the use of operating voltages that are 300–500 V or even higher. These vias are filled by copper electroplating. Prior work on 3D interconnect for integrated circuits (ICs) has mainly focused on through-silicon vias [14]–[17] for low-voltage devices. Through-glass vias have been investigated as means for providing better insulation and low-cost 3D packaging of ICs [18], [19]. More details of our approach to TGV fabrication are described in Section IV.

The interelectrode spacing and the thickness of the Si diaphragm are the most critical dimensional parameters of the design. The interelectrode spacing, the fill-gas, and the interior pressure of the chamber determine the discharge initiation (breakdown) voltage. The breakdown voltage between plane-parallel electrodes is given by the Paschen curve (Fig. 2) [20]:

$$V_b = \frac{Bpd}{\ln Apd - \ln[\ln(1 + 1/\gamma_{se})]}.$$
 (1)

Here *p* is the fill-gas pressure; *d* is the effective length of the breakdown path, approximated by the spacing between the electrodes. *A* (cm<sup>-1</sup>Torr<sup>-1</sup>) and *B* (V-cm<sup>-1</sup>Torr<sup>-1</sup>) are parameters obtained by fitting the first Townsend coefficient,  $\alpha$  (cm<sup>-1</sup>) as a function of E/p (Electric field/gas pressure),  $\alpha = Ap \cdot \exp(-(E/p)/Bp)$ . *A* and *B* depend on the type of gas



Fig. 3. Simulation results of diaphragm deflection at cathode 2 and maximum induced stress for a 2  $\mu$ m, 5  $\mu$ m, and 10  $\mu$ m thick Si diaphragms.

but not on the metal of the electrodes. The parameter  $\gamma_{se}$  is the secondary electron emission coefficient by ion bombardment, which depends on the metal selected for the electrodes and the gas. A larger  $\gamma_{se}$  lowers the breakdown voltage, which is one of the reasons that Ni is used for the electrodes. For each choice of fill-gas in the microdischarge chamber, there is a  $p \cdot d$ product that corresponds to a minimum voltage for discharge initiation. Operating the device near the minimum of Paschen's curve also has the additional benefit that the energy dissipated by the discharge is reduced. For this work Ar at 1 atm. is chosen for the fill-gas in the microdischarge chamber, because it offers lower operating voltages compared to nitrogen and is cost-effective. Based on this choice of gas and pressure, the AK1 spacing is selected as 10  $\mu$ m. The AK2 spacing can be tailored by adjusting the thickness of the bond ring; it is set at 10  $\mu$ m and 30  $\mu$ m in the plasma model and at 30  $\mu$ m in the experiments.

The thickness of Si diaphragm has a significant influence on both sensitivity and dynamic range. A thinner diaphragm offers a high sensitivity but limits the pressure. To determine the appropriate design choice, FEA is performed using COMSOL<sup>®</sup> for pressures up to 50 MPa (Fig. 3). A 5  $\mu$ m thick diaphragm supports a large pressure dynamic range while allowing significant deflection. The deflection of the diaphragm supporting the *K2* electrode is up to about 5  $\mu$ m (0.12  $\mu$ m/MPa) within the fracture limit, which is a significant fraction of the initial *AK2* spacing.

It should be noted that even for the smallest nominal AK2 gap (10  $\mu$ m), a full scale diaphragm deflection of 5  $\mu$ m will only change the pressure within the microdischarge chamber from 1 atm. to about 1.2 atm. This is provided by the following formulae [21], assuming that the ideal gas law is applicable and the gas inside the chamber is isothermal:

$$\Delta d = \frac{3 \cdot \Delta P (1 - v^2) a^4}{16Eh^3}$$
(2)

$$\Delta V = \frac{\pi \cdot a^2 \cdot \Delta d}{3} \tag{3}$$

where  $\Delta d$  is the deflection at the center of a circular diaphragm,  $\Delta P$  is the pressure difference across the

diaphragm, a is the radius, h is the thickness, v is Poisson's ratio of the material, E is Young's modulus and  $\Delta V$  is the volume change due to deflection.

The exterior dimensions of the device are shown in Fig. 1. The total volume of the sensor in the design is 0.05 mm<sup>3</sup>, whereas the microdischarge chamber is only  $2.2 \times 10^{-4}$  mm<sup>3</sup>

# III. MICRODISCHARGE MODELING

First principles computer modeling of the microdischarge pressure sensor was performed to provide insights to the physical processes occurring in such devices and to provide guidance in scaling the devices to smaller dimensions. This effort benefited from the availability of a customized 2-dimensional multi-fluid hydrodynamic computational platform, *nonPDPSIM* developed and refined over a number of years by three of the co-authors [22], [23]. It solves transport equations for all charged and neutral species in the plasma coincident with Poisson's equation for the electric potential and radiation transport. The fundamental equations for charged species are:

$$\nabla \left( \varepsilon \nabla \Phi \right) = -\left( \sum_{j} q_{j} N_{j} + \rho_{s} \right) \tag{4}$$

$$\frac{\partial N_j}{\partial t} = -\nabla \cdot \vec{\Gamma}_j + S_j \tag{5}$$

$$\frac{\partial \rho_s}{\partial t} = \left[ \sum_j q_j \left( -\nabla \vec{\Gamma}_j + S_j \right) - \nabla \left( \sigma \left( -\nabla \Phi \right) \right) \right]$$
(6)

where  $\varepsilon$ ,  $\Phi$ ,  $\rho_s$  and  $\sigma$  are the permittivity, electric potential, surface charge density, and conductivity of solid materials; and for species j, the terms  $N_i$ ,  $\vec{\Gamma}_i$ ,  $S_i$ , and  $q_i$  represent charged species number density, species flux, source function, and charge, respectively. Poisson's equation [Eq. (4)], the transport equations for conservation of the charged species j [Eq. (5)], and the material and surface charge balance equation [Eq. (6)] are simultaneously integrated using a sparse-matrix and Newton iteration technique. Poisson's and charge density equations are solved throughout the computational domain to address electric field penetration into dielectrics. Electrically floating metal features are approximated as dielectrics having sufficiently high conductivities such that the floating feature is equipotential, and there is essentially no internal electric field within it. The electron energy equation is integrated for average energy  $E_{ave}$ ,

$$\frac{\partial (n_e E_{ave})}{\partial t} = \vec{j} \cdot \vec{E} - \nabla \cdot \left(\frac{5}{2}\vec{\Gamma}_e E_{ave} - \lambda \nabla T_e\right) - n_e \sum_i \Delta E_i \kappa_i N_i \qquad (7)$$

where  $T_e$  is the electron temperature defined as  $(2E_{ave}/3)$ ,  $n_e$  is the electron density,  $\kappa_i$  is the rate coefficient for collision process *i* with species having density  $N_i$  and energy loss  $\Delta E_i$ ,  $\lambda$  is the electron thermal conductivity, and  $\Gamma_e$  is the electron flux. Transport and rate coefficients are obtained by solving Boltzmann's equation for the electron energy distribution, and constructing a table of coefficients as a function of  $T_e$ . This table is then interpolated and updated during execution



Fig. 4. Schematics of the two geometries of the microdischarge pressure sensor modeled in this effort. (a) The microdischarge chamber for a fully assembled sensor having a 10  $\mu$ m gap. The microdischarge chamber has the vertical scale expanded by a factor of 3.3 for clarity. (b) The tested device with spacers. The vertical scale is expanded by a factor of 2.

of the model. These electrons are referred to as bulk electrons. Secondary electrons emitted from surfaces and which are accelerated in the sheaths adjacent to surfaces are referred to as beam electrons. These electrons are tracked using a Monte Carlo simulation. Secondary electrons are emitted from surfaces by bombardment by fluxes of ions and UV photons.

The computational platform, *nonPDPSIM* uses an unstructured mesh which enables fine features of the actual devices to be modeled over a dynamic range of  $10^4$ , and so an assessment of narrowly spaced elements and small features can be made. The finest resolution of the mesh for cases discussed here is 0.6-0.8  $\mu$ m. Although the mesh is static – that is, it does not evolve during the calculation – the change in differential current as a function of pressure can be modeled with a series of calculations, each with a different deformation of the diaphragm. Extensive databases are available for plasma and neutral reactions occurring in the Ar-filled microdischarge chamber [22].

Schematics of the two geometries of the microdischarge pressure sensor modeled here are in Fig. 4. The microdischarge chamber is shown in Fig. 4(a) for a sensor having a 10  $\mu$ m gap, with the vertical scale expanded by a factor of 3.3 for clarity. The glass has a dielectric constant of  $\varepsilon/\varepsilon_0 = 3.9$ . The reference cathode (*K1*) and the anode (*A*) are exposed to the microdischarge chamber with electrodes 1  $\mu$ m in thickness and 35  $\mu$ m wide, separated by 10  $\mu$ m. The exposed area of the sensing cathode (*K2*) is directly under the anode; elsewhere is it covered with dielectric. This configuration represents the fully assembled device. The device used for experimental validation is shown in Fig. 4(b) (with the vertical scale expanded by a factor of 2).

The simulated electron density and ionization source by bulk and secondary beam electrons for the device with the

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Fig. 5. Plasma properties for the microdischarge sensor having a 10  $\mu$ m gap operating in 1 atm. of Ar with 400 V applied to the anode as a function of time after application of voltage. (a) Electron density and (b) ionization source (the sum of ionization by bulk and secondary beam electrons). The flood contours are on a log scale over 4 decades. The maximum value in each frame is noted. (The vertical scale is expanded by a factor of 3.3).

10  $\mu$ m gap are shown in Fig. 5. The ionization source is the volumetric rate at which electrons and ions are produced by collisions between electrons and atoms. The fill-gas is 1 atm. (760 Torr) of Ar. For this simulation, the applied voltage on the anode is 400 V, while the cathodes are grounded. Ballast resistors of 100  $\Omega$ , 100  $\Omega$  and 500  $\Omega$  are assumed, in series with K1, K2, and A, respectively. The plasma is initiated by a small amount of electric field emission of electrons from the edges of K1 and K2 resulting in a negligibly small current density of  $10^{-2}$  A-cm<sup>-2</sup>. These electrons rapidly avalanche in the geometrically enhanced electric fields at the edges of the anode. The electron density increases from  $10^9$  to nearly  $10^{16}$  cm<sup>-3</sup> over a period of 10 ns, creating a conductive plasma in front of the anode. This conductive plasma then reduces the electric field around the anode, translating the large electric field to the periphery of the plasma and toward the cathodes. When the plasma reaches K2, ion and photon bombardment of its surface produces secondary electrons, which are accelerated back into the plasma and maintain a large rate of ionization. The large electric field directed towards K1 enables the plasma to spread in that direction, eventually covering K1 after about 20 ns. Bulk electrons are prevented from reaching K1 and K2 by the electrodes' large negative potential with respect to the plasma, as shown by the gaps in bulk electron density adjacent to K1 and K2 in Fig. 5(a). Current to these electrodes is provided by the conduction current of ions and displacement current.

The simulated electron density and ionization source in the device with a 30  $\mu$ m gap are shown in Fig. 6. Ballast resistors of 1000  $\Omega$ , 1000  $\Omega$  and 20 M $\Omega$  are assumed, in series with

Fig. 6. Plasma properties for the tested microdischarge sensor having a 30  $\mu$ m gap operating at 770 Torr of Ar with 480 V applied to the anode as a function of time after application of voltage. (a) Electron density and (b) ionization source (the sum of ionization by bulk and secondary beam electrons). The flood contours are on a log scale over 5 decades. The maximum value in each frame is noted. The vertical scale is expanded by a factor of 2.

K1, K2, and A, respectively. The pressure is 770 Torr (consistent with the experiments) and the initiating current density from the cathodes is  $10^{-4}$  A-cm<sup>-2</sup>. The voltage on the anode is assumed to be 480 V with a rise time of 5 ns. Although the trends are similar to the smaller AK2 gap device, there are qualitative differences due to the significantly larger gap and ballast resistor. The closer proximity of cathode K1 to the anode A initiates the plasma first between these two electrodes. The larger gap produces a lower electric field between K2 and anode A, and so the relative contribution of electric field enhancement to ionization at the edges of the electrodes is greater. The larger ballast resistor produces a larger decrease in voltage across the plasma, since as current is collected there is a larger voltage drop across the ballast resistor. This reduced voltage, combined with the larger gap, produces a lower electric field in the bulk plasma and a lower electron density. A peak value of 10<sup>14</sup> cm<sup>-3</sup> is observed compared to nearly  $10^{16}$  cm<sup>-3</sup> for the other compact device.

### IV. FABRICATION

The fabrication process requires 6 masks, 3 for glass processing (Fig. 7) and 3 for silicon processing (Fig. 8). The glass processing includes the laser-drilling of the TGVs, followed by the filling of the vias using copper electroplating. The next steps include the patterning of the contact pads on the exterior side of the wafer and the indium bond ring on the interior side of the discharge chamber. Finally, the Ni electrodes are patterned inside the microdischarge chamber. The silicon processing includes the deposition and patterning







Fig. 7. Process sequence for the glass wafer. 1) Through-holes are laser drilled. 2) Glass wafer is attached to a dummy Si wafer by eutectic bonding. A 4–6  $\mu$ m thick electroplated indium layer on the dummy wafer serves as the seed layer for Cu electroplating. 3) Through-holes are filled by Cu electroplating. 4) A lapping step removes excess Cu and the dummy wafer. 5) The contact pads are deposited and patterned on the exterior of the glass wafer. 6) The bond ring and *K2* contact are defined on the interior side of the glass wafer. The In bond ring is covered with 50 nm Au for protection. 7) Electrodes are deposited on the interior of the glass wafer.

of an insulating oxide on the Si device layer of a siliconon-insulator (SOI) wafer. This is followed by the patterning of the Au bond ring and K2 electrode. Next, the glass and silicon chips are aligned and attached using a Au-In eutectic bonding method. Post-bonding, the Si diaphragm is released from the handle wafer by a deep reactive ion etching (DRIE) process, using the buried oxide layer as the etch stop.

#### A. Glass Processing

The glass processing uses 300  $\mu$ m-thick Schott Borofloat<sup>®</sup> glass wafers. In order to provide electrical contact from the



## 4. Release handle wafer

Fig. 8. Process sequence for the Si wafer. 1) Silicon dioxide layer is grown and patterned on SOI for diaphragm insulation, and K2 contact is then defined. 2) Au bond ring and K2 contact on the interior of SOI wafer are electroplated. K2 electrode is deposited by a lift-off process. 3) Eutectic bonding is performed between glass and silicon wafers. 4) Handle wafer is released by a backside dry etching.

pressure sensor electrodes (located within the sealed chamber) to the contact pads (located outside the chamber), vias are drilled (Precision Microfab, Severna Park, MD) using a 193 nm ArF excimer laser. This machining process has a depth control of approximately  $\pm 5 \ \mu$ m, a lateral precision of 1-2 $\mu$ m, and a profile taper of 88.1°. The actual machined holes are 47.5  $\mu$ m on the exterior side and 15.8  $\mu$ m on the interior side for a machining profile of 87° [Fig. 9(a)].

A variety of methods can be used for achieving an electrical connection through the glass vias, including thin-film deposition, packing and melting of solder balls or powder, and electroplating. The high aspect ratios of the TGV structures make it impractical to achieve sufficient sidewall coverage for reliable electrical contacts using thin film deposition. The use of solder particles yields limited success because of inconsistent reflow when heated to the melting temperature (183°C for 37Pb/63Sn) and beyond (up to 280°C). Although the exact cause of this behavior has not been determined, it is possibly related to the large ratio of surface area to volume, which is known to prevent recrystallization.

Electroplating provides the consistency and scalability for filling the TGVs. Although a variety of plating metals are available, In and Cu are the best candidates for this application. Indium has a low reflow temperature (156°C), which allows temperature cycling post-plating in order to remove pinholes or voids. Copper offers lower resistivity and a higher plating rate. The higher re-melting temperature can also accommodate



Fig. 9. Optical photos of fabricated glass chip. (a) Laser drilled TGVs, out-chamber side. (b) TGVs are electroplated with copper; bond ring and K2 contact are patterned with indium and Au protection. (c) Zoom-in view of the microdischarge chamber. (d) Anode and cathode I are patterned with Ti/Ni as electrodes by a lift-off process.

a higher operating temperature for the pressure sensor. Both metals were successfully plated in experiments.

Before electroplating, the glass wafer is attached to a dummy Si wafer coated with the appropriate metal seed layer (e.g. Ti/Au) for electroplating. Maintaining close contact and minimizing movement between the glass wafer and seed layer are critical. To ensure this, first the glass wafer is coated with a thin layer of Ti/Au 30 nm/300 nm whereas the dummy Si wafer is coated with a 30 nm/300 nm Ti/Au layer and a 4–6  $\mu$ m-thick electroplated In layer. Then the dummy Si wafer is bonded to the glass wafer using Au-In eutectic bonding. (Two other attachment options include photoresist to affix the dummy wafer [18] or electroless plating [19].) Following a degassing step to remove bubbles from the vias, Cu plating is performed (Enthone Cuprostar<sup>®</sup> CVF1) at 24°C. Pulse plating with periodic reversal of polarity is used to provide uniform plating across the TGVs on the wafer. The effective current density is 15–20 mA/cm<sup>2</sup>. After the plating, the stacked structure is lapped from the front to remove excessive metal build-up and planarize the surface, and from the back to grind off the dummy Si wafer. The measured resistance of the TGV is  $<5 \Omega$ .

The next processing steps involve patterning the metal contact pads on the exterior side, followed by the patterning of the indium bond ring and Ti/Ni electrodes (A and KI) on the interior of the microdischarge chamber.

The Ti/Au (30 nm/300 nm) contact pads (located on the exterior of the glass chip) are patterned using the lift-off technique. To prevent discharges outside the chamber (i.e., across the contact pad features), the corresponding contacts for the anode and two cathodes are strategically spaced farther apart compared to their spacing inside the chamber (10  $\mu$ m), and partially coated with an insulating layer of epoxy above the TGVs (Fig. 1).

The In bond ring (4  $\mu$ m-thick, coated with a 50 nm-thick Au layer) and one side of *K*2 contact are then formed by evaporation and lift-off (Fig. 9). The Ti/Ni (20 nm/200 nm) electrodes (34.5  $\mu$ m-wide) are formed by lift-off achieving an *AK1* spacing of approximately 11.1  $\mu$ m. As shown in Fig. 9(d), the alignment of the TGVs and Ni electrodes is reasonably good and centered.

## B. Silicon Processing

The silicon processing utilizes SOI wafers with a Si device layer (5  $\mu$ m-thick), a buried silicon dioxide layer (2  $\mu$ m-thick), and a Si handle wafer (500  $\mu$ m-thick). The Si device layer includes As doping for low resistivity (<0.005  $\Omega$ -cm); this is necessary to electrically route the *K*2 electrode to the *K*2 contact. When the glass and Si wafers are bonded, the *K*2 contact on the Si wafer electrically connects to the *K*2 contact on the glass wafer. This connection is then routed through the TGV to the contact pad for *K*2. The buried oxide layer provides a well-defined etch stop, which can later facilitate the final diaphragm release of the Si device layer by a backside dry etch of the handle wafer.

Silicon dioxide is grown (100 nm-thick, by dry oxidation at 1000°C) and then deposited (900 nm-thick low temperature oxide) for a total thickness of 1  $\mu$ m on the Si device layer to provide electrical isolation of the bond ring from the *K*2 electrode and contact. The oxide is patterned using a dry etch process based on CHF<sub>3</sub> and CF<sub>4</sub> to expose the doped device layer for the *K*2 contact. Next, an 8  $\mu$ m-thick Au bond ring and *K*2 contact are provided by electroplating. In a following step, the oxide is removed in the region of the *K*2 electrode, which is then formed by sputtering and lift-off of Ti/Ni (20 nm/200 nm).

#### C. Eutectic Bonding

The transient liquid phase bonding technique has been used for vacuum packaging [24], [25] and wafer level attachment of ceramics (e.g., PZT) to Si [26]. The motivations include: (1) a relatively low initial melting temperature ( $200^{\circ}$ C) for the bonding step with the potential for high re-melting temperatures e.g., 500°C with Au-In; (2) the ability to bond a variety of surface profiles; and (3) the ability to control the thickness of the bond layer. The Au-In system was first investigated as a fluxless soldering technique in electrical packaging [27], [28]. There are two basic types of Au-In bonding systems depending on the percent weight of the two metals: indium-rich [27] and gold-rich bonding [28]. If the weight percentage of indium is higher than 54%, also called "indium-rich", the alloy is a mixture of indium and AuIn<sub>2</sub>, which means the re-melting (or de-bonding) temperature is still 156°C. If the weight percentage of indium is between 36.8–54 wt.%, the alloy is a mixture of AuIn and AuIn<sub>2</sub> intermetallic compounds with a re-melting temperature of 495.4°C and is considered "gold-rich".

For the pressure sensor described in this paper, Au-rich bonding is used to ensure a relatively high operating temperature ( $\approx$ 500°C). This bonding process does not depend on



Fig. 10. Chip level eutectic bonding of Au-In bond rings. (a) Glass and silicon chips, before and (b) after bonding, against a U.S. penny ( $\emptyset$ 19.05 mm). (c) Zoom-in view of one bonded ring, viewed through glass. (d) SEM image of the cross-section of the Au-In bond ring. (e) SEM details of the bond structure. (f) Electron dispersive spectroscopy shows the diffusion of the Au and In components and the resulting Au and In intermetallic compounds that have formed.



Fig. 11. Photograph of the final assembly of the glass and silicon chips, viewed from the exterior of the glass.

substrate materials (either glass or silicon). In this experimental investigation both cases, with Au on glass/In on silicon, and Au on silicon/In on glass were studied and successfully bonded. The bonding was performed in a vacuum oven at 200°C with an applied pressure >1 MPa for 90–120 mins. Fig. 10(d) shows the cross-section of the bond ring structure captured using scanning electron microscopy (SEM). Electron dispersive spectroscopy (EDS) was used to evaluate the composition of the bond ring [Fig. 10(f)]. The EDS shows diffusion of the Au and In layers that form the intermetallic compounds.

## V. EXPERIMENTAL RESULTS

To evaluate the impact of multiple conditions for the interior of the microdischarge chamber, a test structure was formed in which the SOI wafer was thinned to 100  $\mu$ m, but the handle wafer was not completely removed (Fig. 11). The electrical characterization was performed with a glass chip



Fig. 12. Test setup for the glass and Si chip assembly. A micromanipulator applies force at the center of the 100  $\mu$ m-thick Si diaphragm to deflect the Si chip and change the *AK2* spacing.



Fig. 13. Representative waveforms of the microdischarges collected at the cathodes obtained from an oscilloscope during the tests.

and a SOI chip that were held together with a porous epoxy bond instead of an eutectic bond. The test structure was left unsealed and tested in an argon environment. The experimental setup is illustrated in Fig. 12. A piezoelectric actuator was used to apply a force at the center of the assembled chip to induce a diaphragm deflection, which emulates a large external pressure. Voltage pulses of 1 ms duration were applied to the anode. Multiple microdischarge pulses were produced during each voltage pulse. Ballast resistor values of 10 M $\Omega$  and 20 M $\Omega$  were utilized in separate sets of experiments, while the currents going through two competing cathodes *I1* and *I2*, were captured as voltages across 1 k $\Omega$  resistors.

A representative waveform of a single microdischarge pulse is shown in Fig. 13. The typical duration is several hundred nanoseconds with decaying oscillation. Possible sources of parasitic capacitance, to which the oscillation may be attributed, include the oscilloscope probes connected to K1and K2. When a voltage pulse is applied to the anode, it also charges the parasitic capacitance on the anode, which can potentially contribute to the peak transient discharge currents.

The relation between differential current and estimated equivalent pressure is plotted in Fig. 14. The force applied



Fig. 14. Differential currents from test results versus estimated equivalent pressure and inter-electrode gap from simulation results based on a 5  $\mu$ m thick Si diaphragm. *II* and *I2* are peak values of envelope curves for *AK1* and *AK2* discharge waveforms, respectively. Tests are in Ar with the microdischarge chamber interior pressure between 650 Torr and 770 Torr. Applied voltage pulses are 480 V to 580 V. Ballast resistors are 10 M $\Omega$  and 20 M $\Omega$ . Every point is the average of 5–8 measurements. Error bars indicate one standard error.

to the diaphragm was converted to equivalent pressure using FEA. In four sets of experiments, the impacts of chamber pressure, applied voltage, and ballast resistor were investigated. When the ballast resistor value increases, the nature of the microdischarges changes and impacts the distribution of the cathode currents. For a chamber pressure of 770 Torr and using a 20 M $\Omega$  ballast resistor, the operating voltage was 480 V and the differential currents ranged from -0.35 to -0.5. With a smaller 10 M $\Omega$  ballast resistor, the differential currents uniformly decreased.

Although the device is intended to operate with the microdischarge chamber at about 1 atm., the impact of lower pressure was also evaluated (Fig. 14) with the test structure. At an interior pressure of 650 Torr, the (fractional) current in K1 was lowered, as expected from the increase in the mean free path. Additionally, the operating voltage increased from 480 V for 770 Torr to 580 V for 650 Torr; this was also expected, given the nature of the Paschen curve for Ar.

The impact of operating voltage on the discharge characteristics is also evident in Fig. 14. For the microdischarge chamber pressure of 770 Torr and a ballast resistor of 10 M $\Omega$ , two magnitudes of voltage pulses were evaluated: 480 V and 500 V. For the 480 V pulses, the resulting differential current distribution as a function of diaphragm deflection does not indicate a clear trend. However, at 500 V, the impact of relative deflection can be clearly observed in Fig. 14. This indicates the existence of a minimum threshold for the operating voltage of the pressure sensor. Thus, the combination of voltage (480 V) and ballast resistor (10 M $\Omega$ ) is not expected to be selected for use.

Based on the electrical results, the mechanical load changed AK2 by approximately 5  $\mu$ m. This corresponds to the deflection expected from a 5  $\mu$ m thick diaphragm of oval shape under an external pressure of about 40 MPa as noted in Section II B, and in Fig. 3.

#### VI. DISCUSSION

The plasma modeling in this work provides insights to the fundamental processes responsible for the differential current, including plasma initiation, evolving and current sustaining. This simulation guides the electrodes design and device scaling to smaller dimension, although no quantitative data directly predict and match experimental results which indeed are affected by multiple experimental conditions.

The device is interrogated using high voltage pulses, and the energy consumption for each pulse is about 5 mJ. For previously reported work on pulsed microdischarge-based devices [8], the energy consumption can be as low as 2-20  $\mu$ J/pulse. The power consumption for device described in this work depends on the duty cycle, but for 1 reading each second, it is ~5 mW.

Looking forward, it is notable that devices incorporating microdischarges are attractive for high temperature operation as electron temperatures are typically many eV (1 eV=11,600 K) and so are not significantly perturbed by a high ambient temperature [29], [30]. For the conditions encountered in these devices, ions have temperatures moderately above ambient with transport coefficients that are also not particularly sensitive to high operating temperatures. Microdischarge-based pressure sensors have been operated as high as 1000°C [3]. Other pressure sensors for high temperature utilize Fabry-Perot and other interferometers [1], and piezoresistors in high band gap materials such as SiC (up to 600°C) [31] and even Si (up to 600°C) [32]. The temperature tolerance of monolithic microdischarge-based pressure sensors will be investigated in our future efforts.

## VII. CONCLUSION

There are a number of conclusions that can be drawn from the effort described. First, the differential cathodes arrangement that was investigated - with anode (A) and reference cathode (K1) on the glass substrate, and the sensing cathode (K2) on the diaphragm – was demonstrated to produce a differential output current that was a function of diaphragm deflection. For an AK1 spacing of 10  $\mu$ m, the fractional differential current changed by approximately 20% as AK2 changed from 30  $\mu$ m to 25  $\mu$ m. The overall device size was  $0.585 \times 0.54 \times 0.2$  mm<sup>3</sup>. It was demonstrated that microdischarges could be initiated at voltages below 500 V in an Ar filled microdischarge chamber at about 1 atm. interior pressure. Further, the peak current levels were at a level of  $\approx 10$  mA, and pulse durations were  $\approx 100$  ns, which permits the discharge to remain relatively energy efficient. First principles computer modeling confirmed the roles of the two cathodes in the operation of the device. The microdischarge is initiated in the AK1 gap and travels to AK2. The two cathodes then compete for current. The modeling also shows that the device can operate over a wide range of AK2, from 10  $\mu$ m to 30  $\mu$ m. Tightly packed, high aspect ratio TGVs can be fabricated by combining laser drilling with electroplating. The electroplating can be performed with the aid of a Si dummy wafer that supports a seed layer; the wafer is eutectically bonded to the underside of the glass substrate, and later removed by lapping.

Although the device design and fabrication approaches may continue to evolve, differential microdischarge currents present a viable approach to sensing diaphragm deflection, and so can be implemented in a variety of devices, such as gas or liquid pressure sensors in harsh environments.

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**Christine Kay Eun** received the B.S. (*summa cum laude*), M.S., and Ph.D. degrees in electrical engineering from the University of Michigan, Ann Arbor, in 2004, 2006, and 2011, respectively. She has been funded through various fellowships including the Rackham Merit Fellowship and a fellowship from Sandia National Laboratories, Albuquerque, NM.



Xin Luo received the B.S. degree in Automotive Engineering from Tsinghua University, Beijing, China, in 2010. He is currently working as a Graduate Student Research Assistant towards the Ph.D. degree in the Department of Mechanical Engineering, University of Michigan, Ann Arbor, with a focus on MicroElectroMechanical Systems. His research interests include microsensors for harsh environments.

EUN et al.: MICRODISCHARGE-BASED MONOLITHIC PRESSURE SENSOR



**Jun-Chieh Wang** is a Ph.D. student in the Department of Electrical Engineering and Computer Science at the University of Michigan (UM), Ann Arbor, MI, USA. He received his B.S. and M.S. in physics from National Cheng Kung University, Tainan, Taiwan, in 2003 and 2005, respectively. Before UM, he worked on theoretical and computational plasma physics in the areas of fusion plasma and ionosphere plasma phenomenon. Since 2009, he has been pursuing a Ph.D. at the University of Michigan in Prof. Mark Kushner's group with a

research focus on low temperature plasma science and technology.



Zhongmin Xiong was a research scientist at University of Michigan whose research focuses on numerical modeling of fluid flow, gas discharge and low temperature plasmas. Prior to joining University of Michigan, he worked as a simulation and modeling engineer for research and product development in Schlumberger Ltd. from 2007 to 2009. He did his post-doctoral research on numerical simulations of tokamak edge plasmas at Lawrence Livermore National Lab from 2004 to 2007. He obtained his Ph.D. in flow physics and computation from Stan-

ford University in 2004.

Mark Kushner, photograph and biography not available at the time of publication.



Yogesh Gianchandani (S'83–M'85–SM'05–F'10) is a Professor at the University of Michigan, Ann Arbor, with a primary appointment in the Electrical Engineering and Computer Science Department, and a courtesy appointment in the Mechanical Engineering Department. He also serves as the Director for the Center for Wireless Integrated MicroSensing and Systems (WIMS<sup>2</sup>).

Dr. Gianchandani's research interests include all aspects of design, fabrication, and packaging of micromachined sensors and actuators

(http://www.eecs.umich.edu/~yogesh/). He has published about 300 papers in journals and conferences, and has about 35 US patents issued or pending. He was a Chief Co-Editor of *Comprehensive Microsystems: Fundamentals, Technology, and Applications,* published in 2008. Dr. Gianchandani has served on the editorial boards and program committees of a number of conferences and journals. From 2007 to 2009 he also served at the National Science Foundation, as the program director for Micro and Nano Systems within the Electrical, Communication, and Cyber Systems Division (ECCS).